

Coupling WinCast to ChemApp for the calculation of final microstructure distribution

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Introduction

Micro+ChemApp
Model

Micro+ChemApp
Results

Casting
Simulations

Summary

Outline

- Introduction
- ChemApp coupled microstructure development model for AZ91 Mg alloy
 - Phase distribution depending on diffusion time
 - Al and Zn segregation in solid depending on diffusion time
 - Latent heat release depending on diffusion time
 - Solid fraction vs Latent heat
- Casting simulations with WinCast
- Summary and Future work

Introduction

Micro+ChemApp
Model

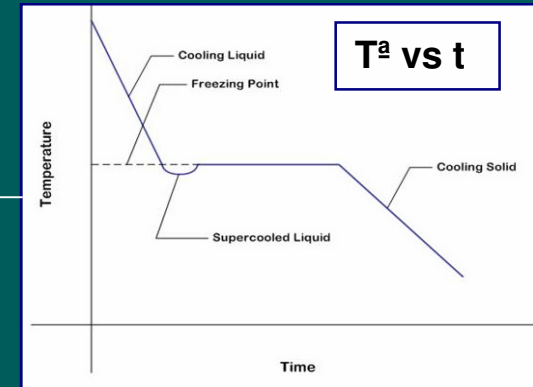
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Micro- model

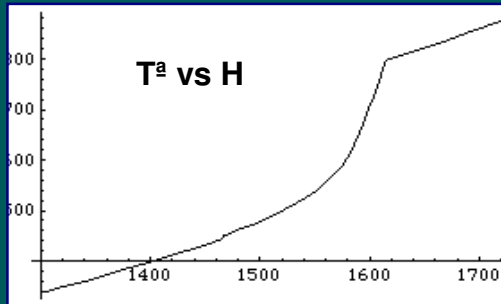
Needs



Gives

CALPHAD

Gives



Needs

Macro- model

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Micro+ChemApp
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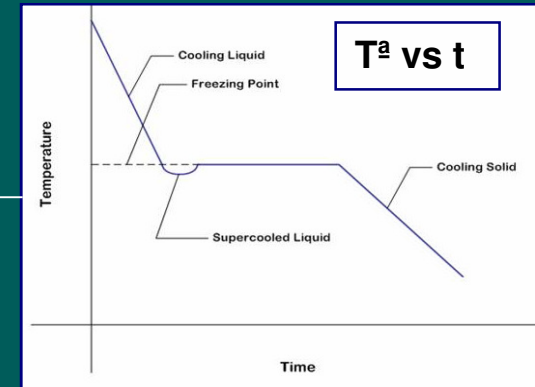
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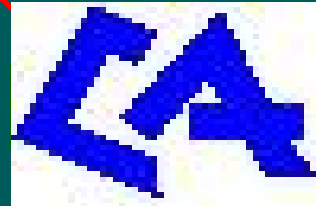
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Micro- model

Needs

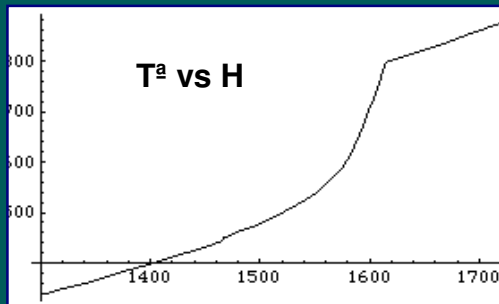


Gives

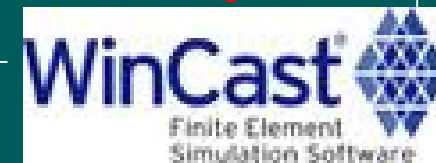


ChemApp

Gives



Needs



Introduction

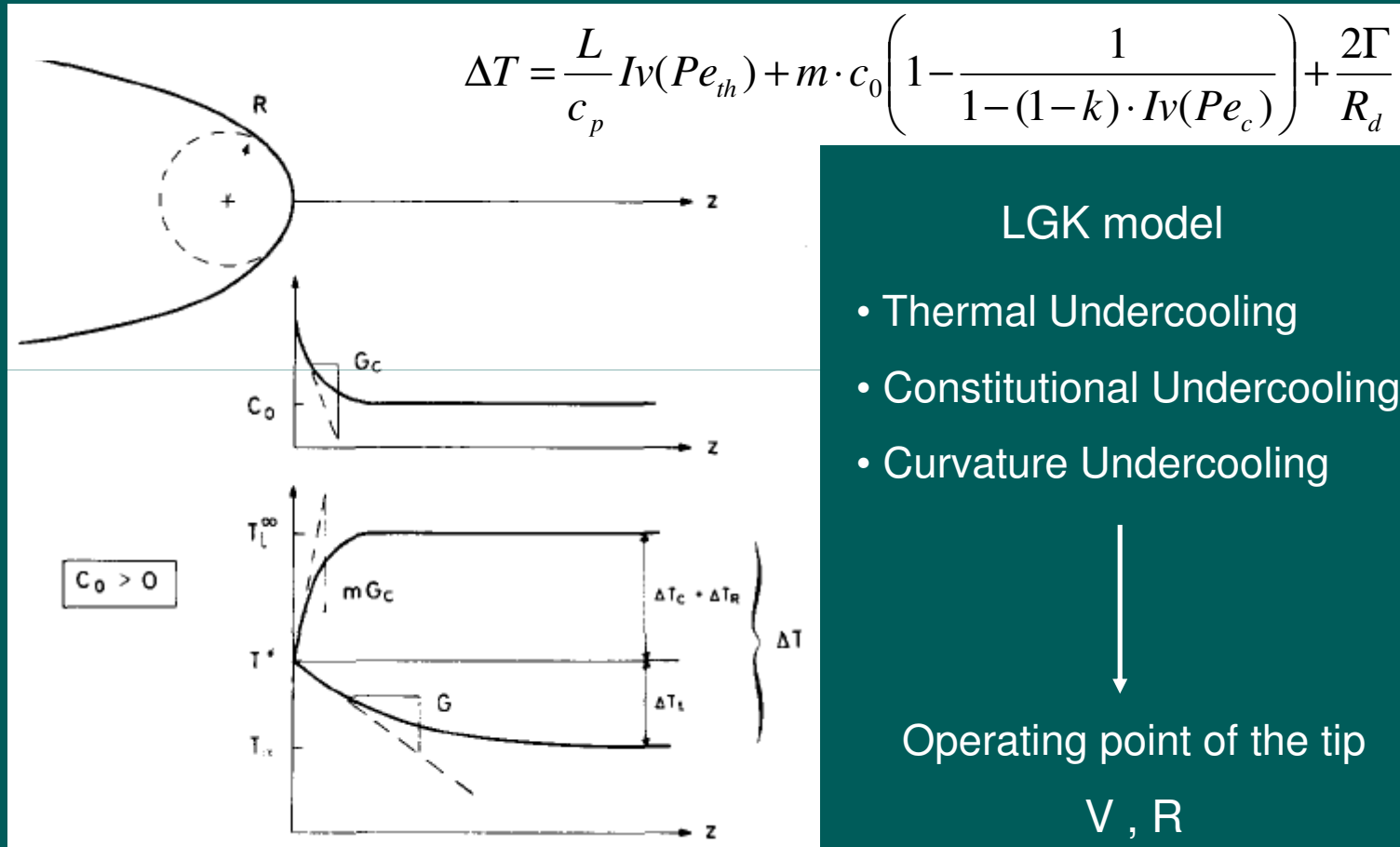
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Dendritic Growth in an Undercooled Melt



LGK model

- Thermal Undercooling
- Constitutional Undercooling
- Curvature Undercooling

Lipton, J., M. E. Glicksman, and W. Kurz. "Dendritic growth into undercooled alloy melts." Mat. Scie. and Eng. 65.1 (1984): 57-63.

Solute Distribution - Diffusion effect in liquid

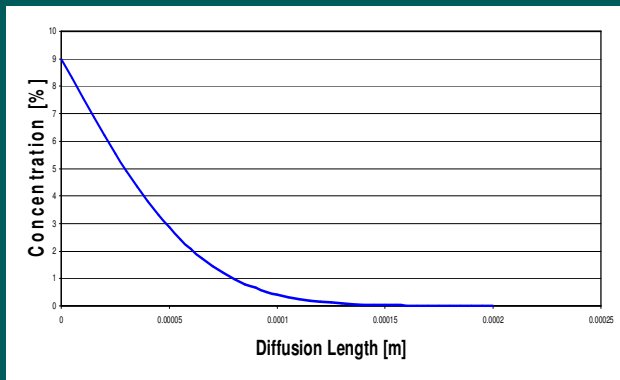
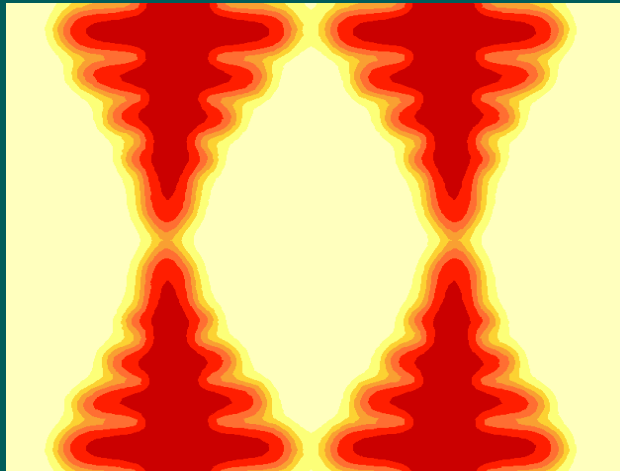
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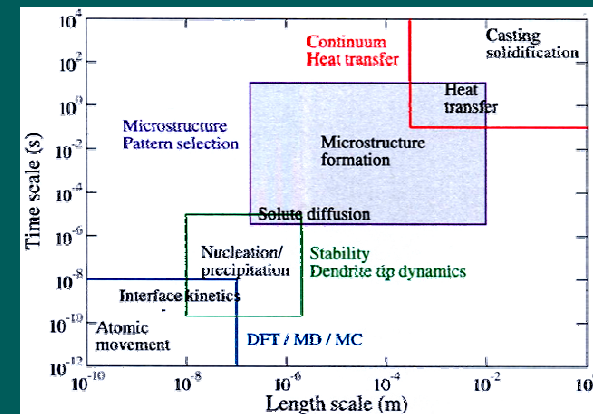
Error function based analytical diffusion equation

$$c = c_0 - c_0 \cdot \operatorname{erf}\left(\frac{x}{\sqrt{D \cdot t}}\right)$$

D → Diffusion coef.

x → Diffusion length

t → Diffusion time



AZ91 Mg alloy (Ternary Mg-Al-Zn)

%	Al	Zn	Mn	Si	Cu	Ni	Other	Mg
AZ91	9	0.42	0.13	0.5	0.1	0.03	max 0.3	Balance

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From 601.75 °C to 433.49 °C

LIQUID -> HCP_A3 (α Mg)

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Results

From 433.49 °C to 430.13 °C

LIQUID -> ALMG_GAMMA (Mg₁₇Al₁₂) + HCP_A3 (α Mg)

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Equilibrium Latent Heat	→	442.63 J/g
Scheil Latent Heat	→	494.43 J/g

Introduction

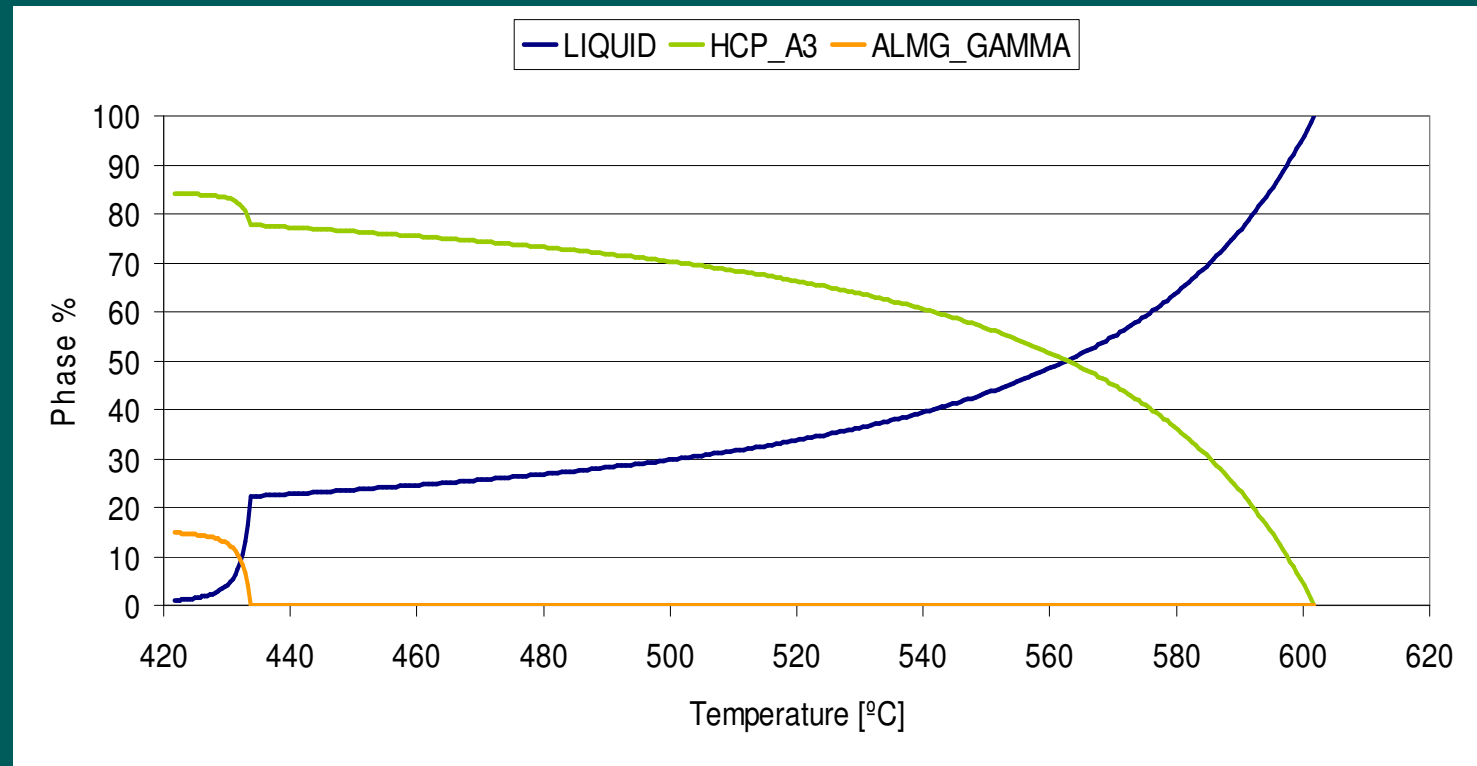
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Phase distribution for a 5 °C/s cooling rate



Diffusion length → 1E-7 m

Diffusion time → 1E-5 s

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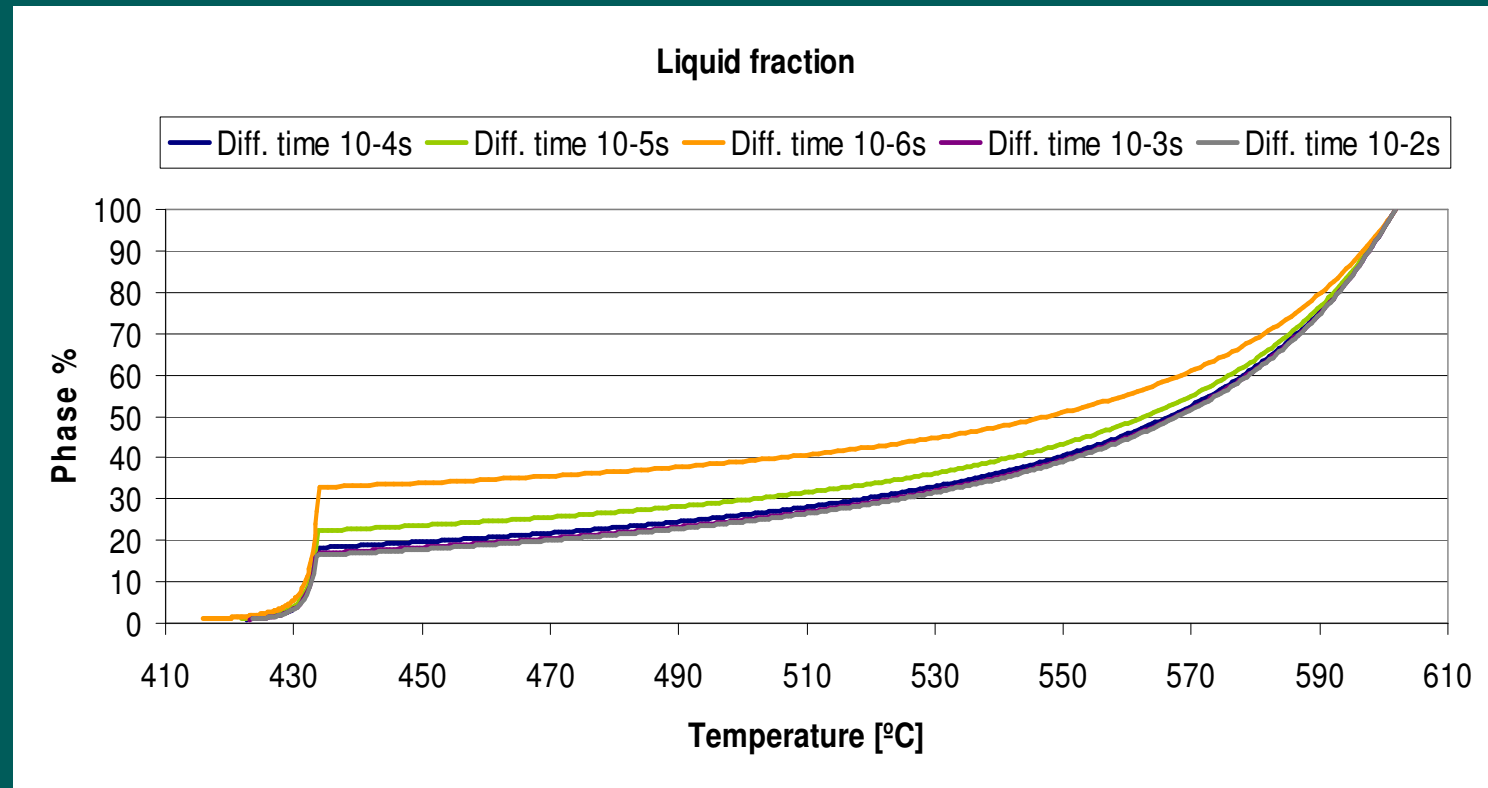
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Liquid Fraction variation depending on the diffusion time



Diffusion length → 1E-7 m

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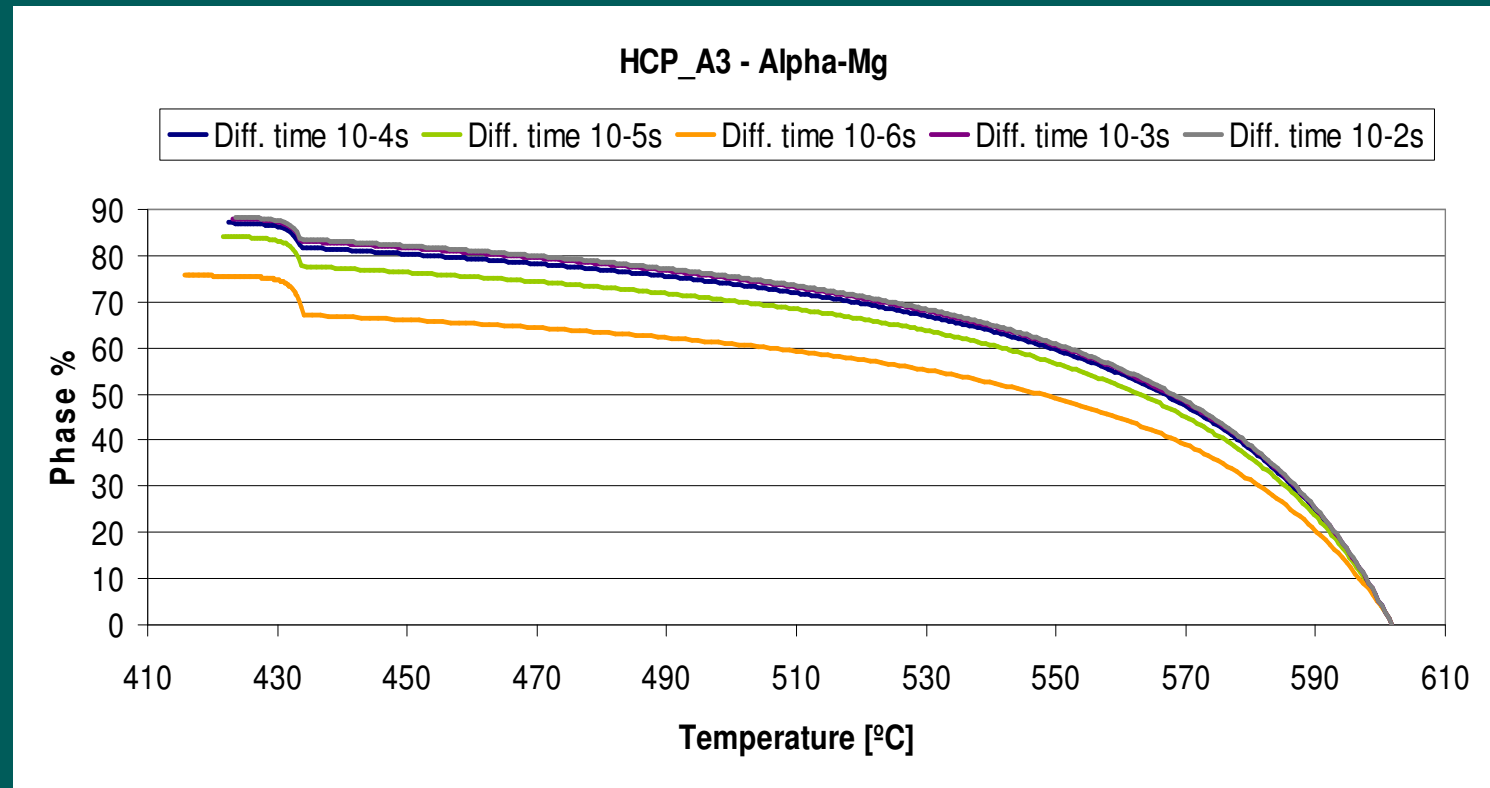
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HCP_A3 (α -Mg) fraction variation depending on the diffusion time



Diffusion length \rightarrow 1E-7 m

Introduction

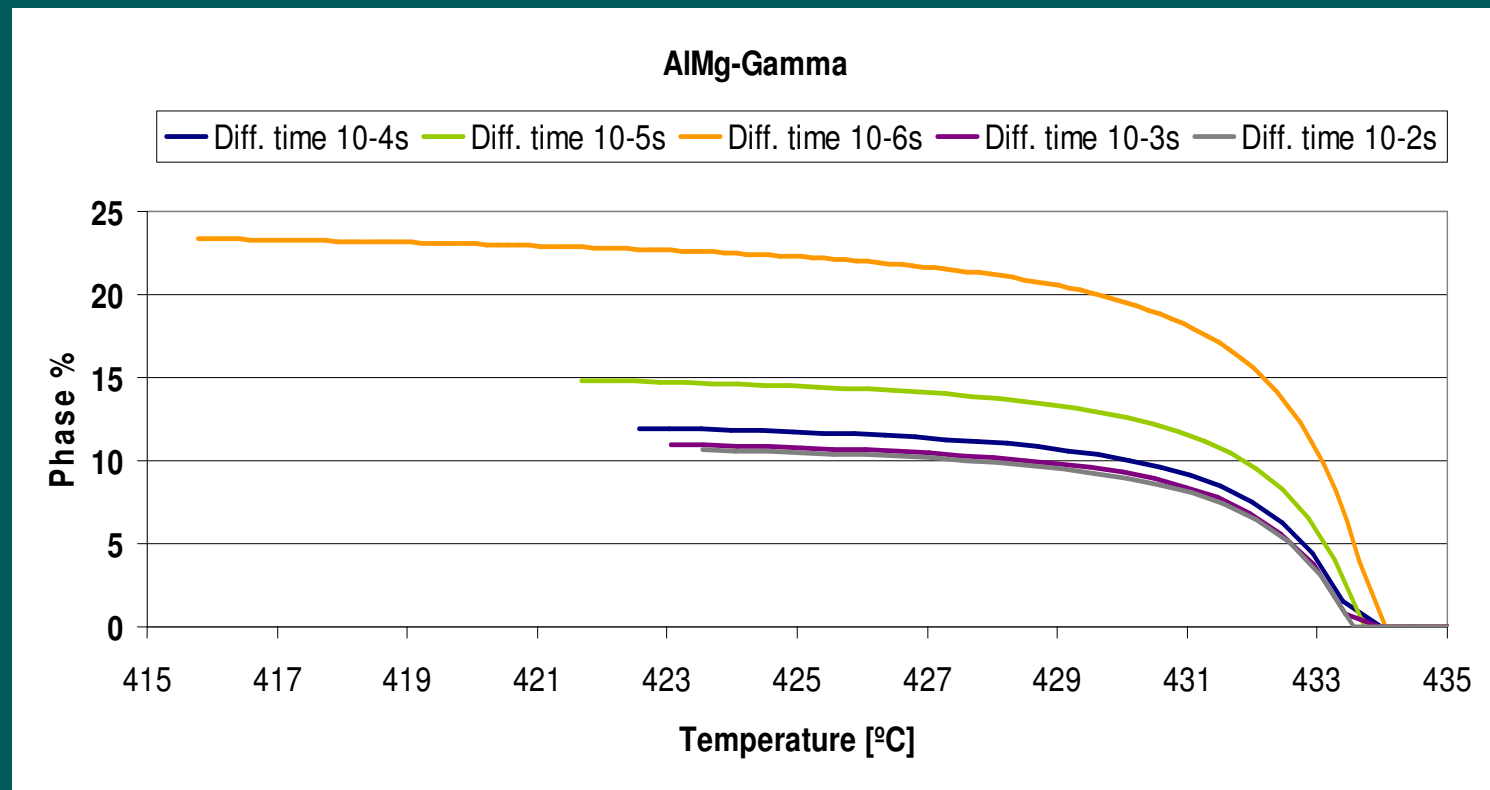
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AlMg-Gamma (Mg₁₂Al₁₇) fraction variation depending on the diffusion time



Diffusion length → 1E-7 m

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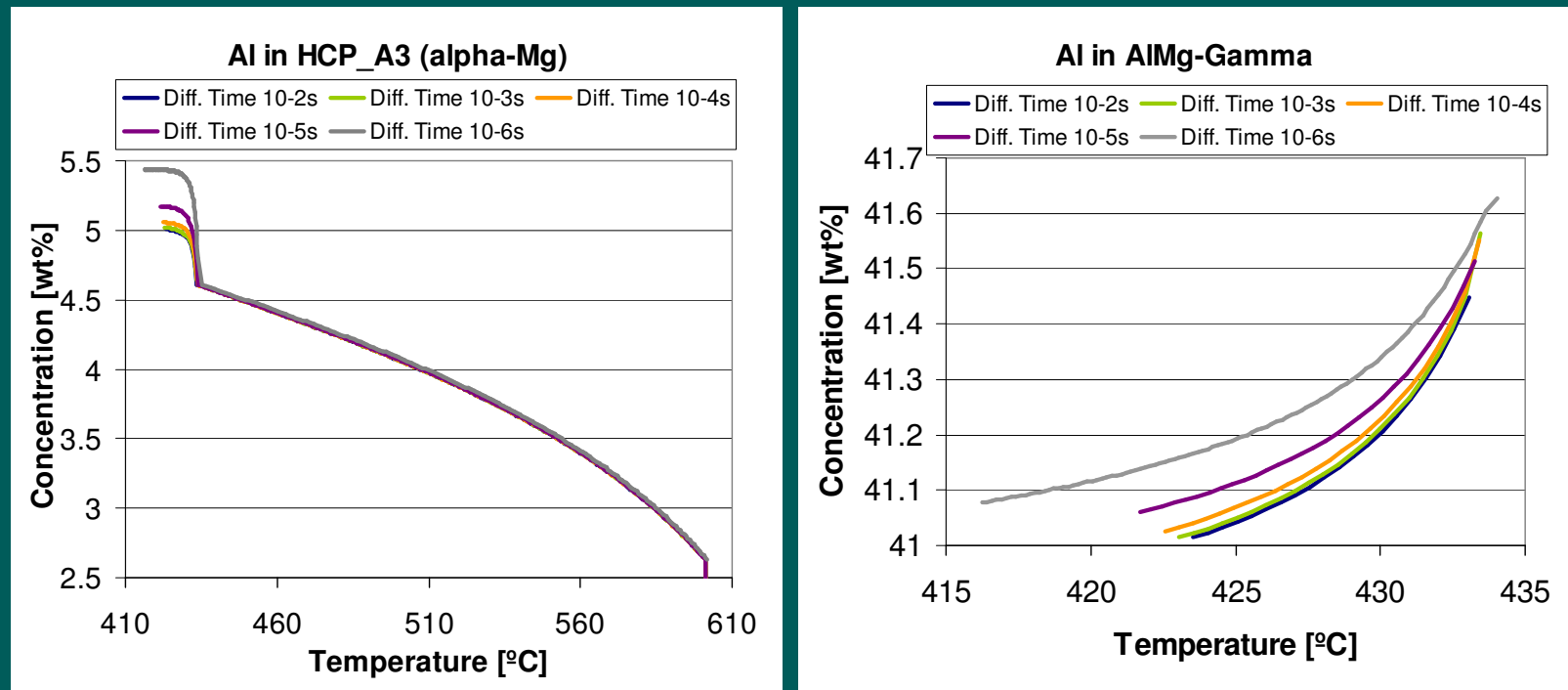
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Al segregation in the solid phases



Diffusion length → 1E-7 m

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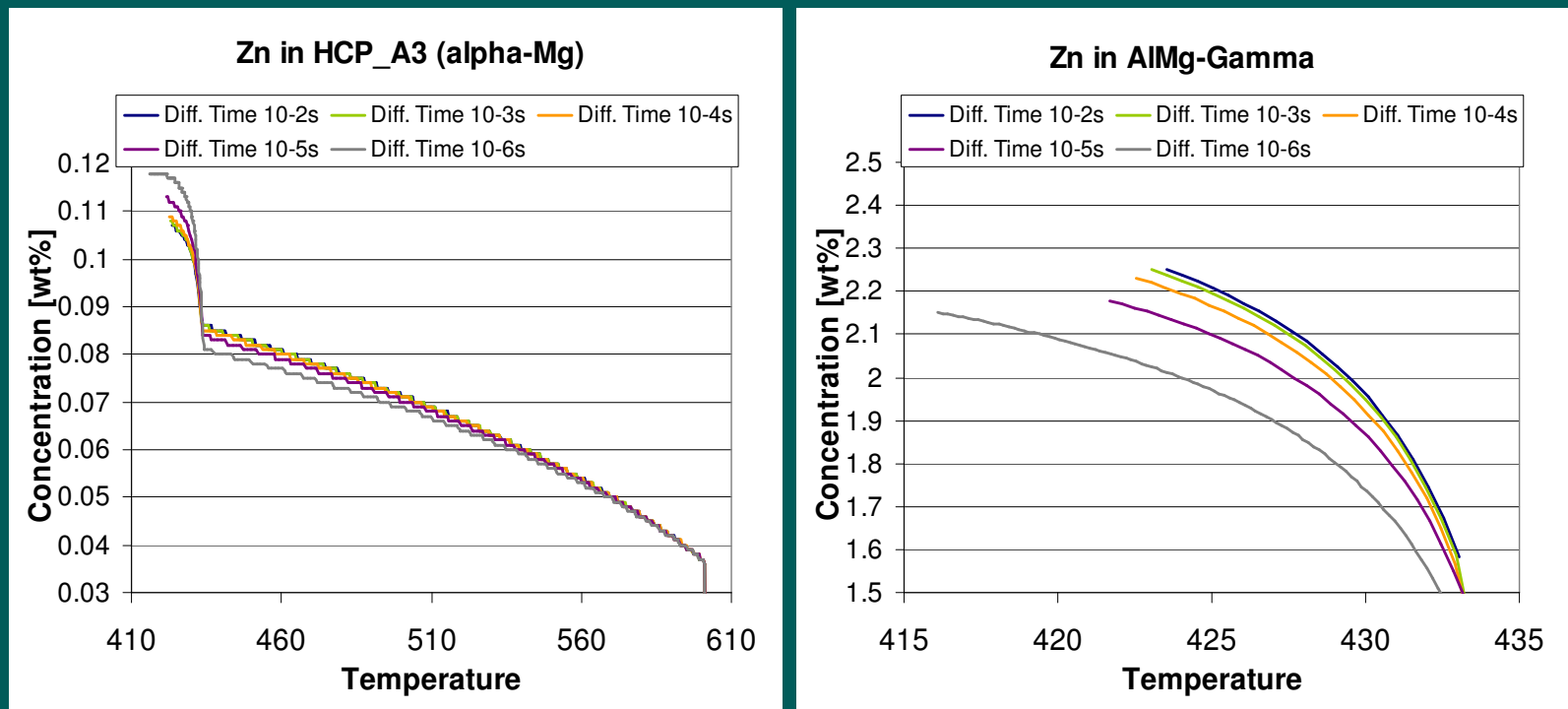
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Zn segregation in the solid phases



Diffusion length → 1E-7 m

Introduction

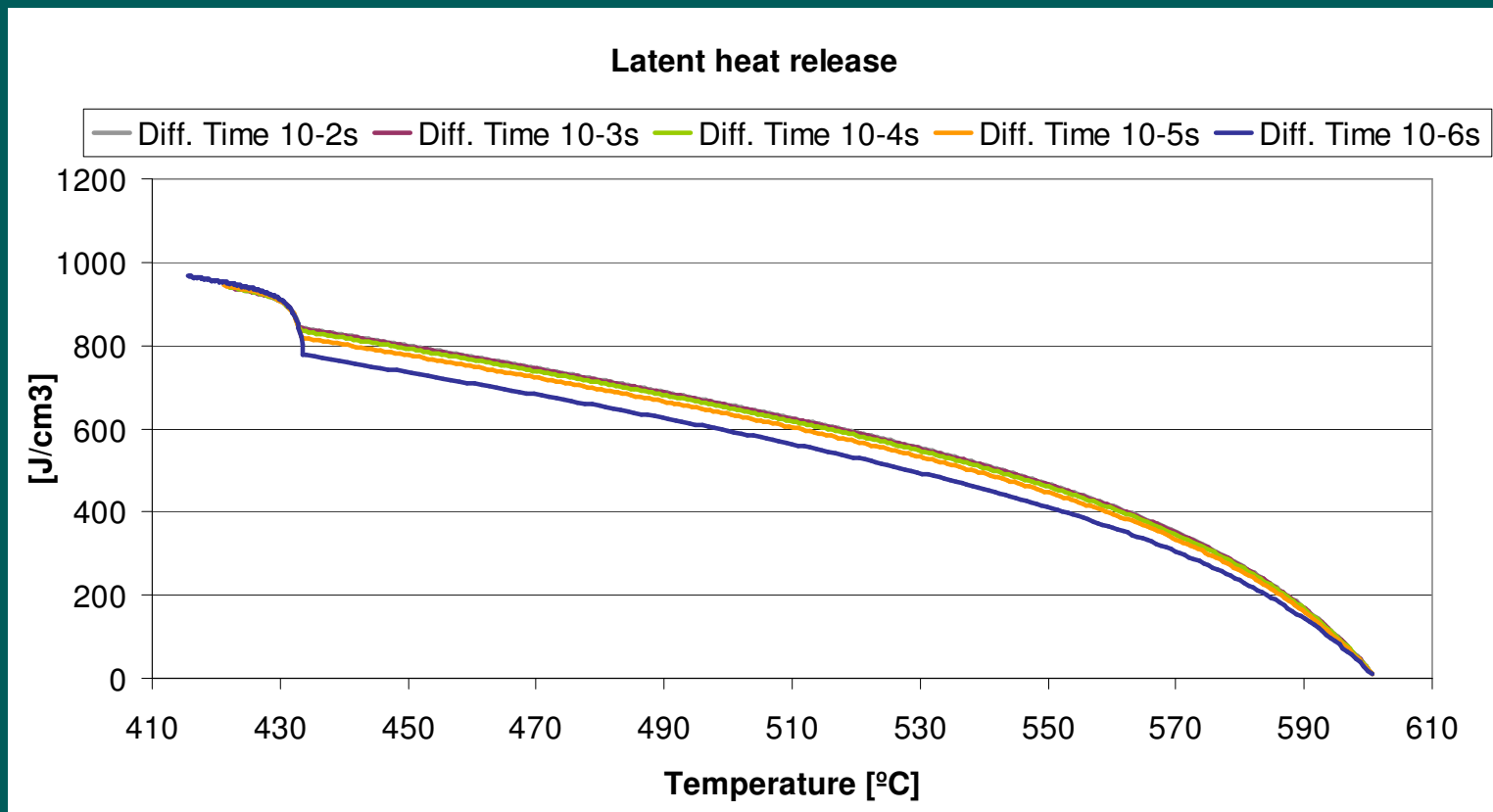
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Latent heat release depending on the diffusion time



Diffusion length $\rightarrow 1E-7$ m

Introduction

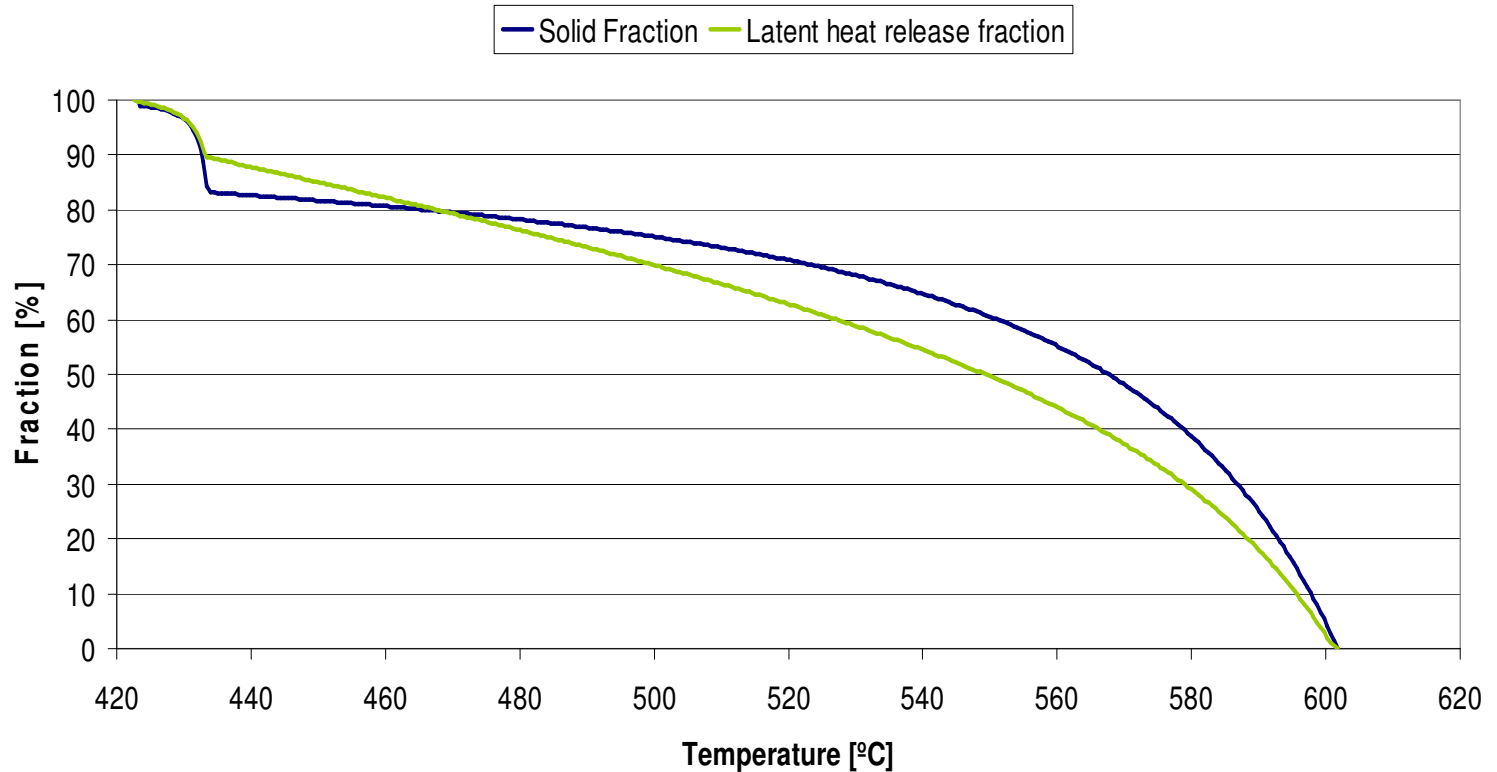
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The solid fraction and the heat release fraction are not proportional



Diffusion length → 1E-7 m

Diffusion time → 1E-3 s



Gravity Die Casting simulations with WinCast

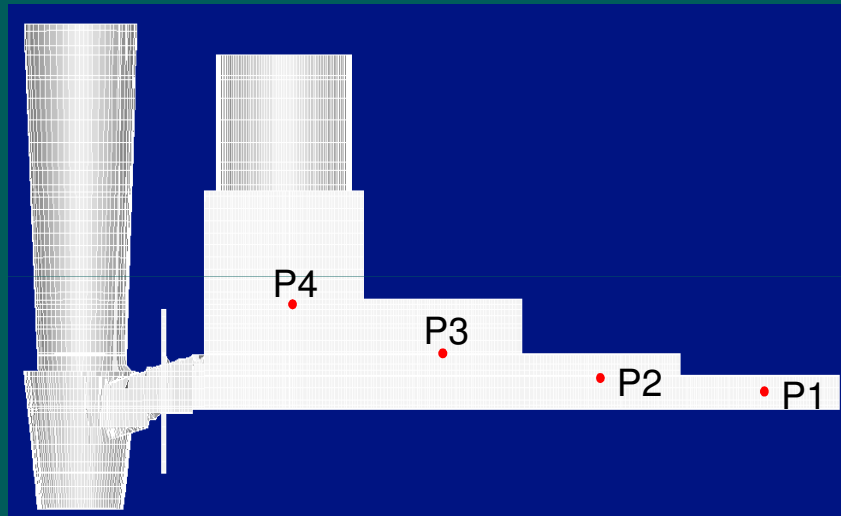
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Casting material: AZ91 Mg alloy

Mould material: GGG40

Melt temperature: 720 °C

Mould temperature: 230 °C

Filling time: 3 s

Part dimensions:

Each step is 70 mm long by 110 mm wide

8, 12, 20 and 40 mm thick respectively

Introduction

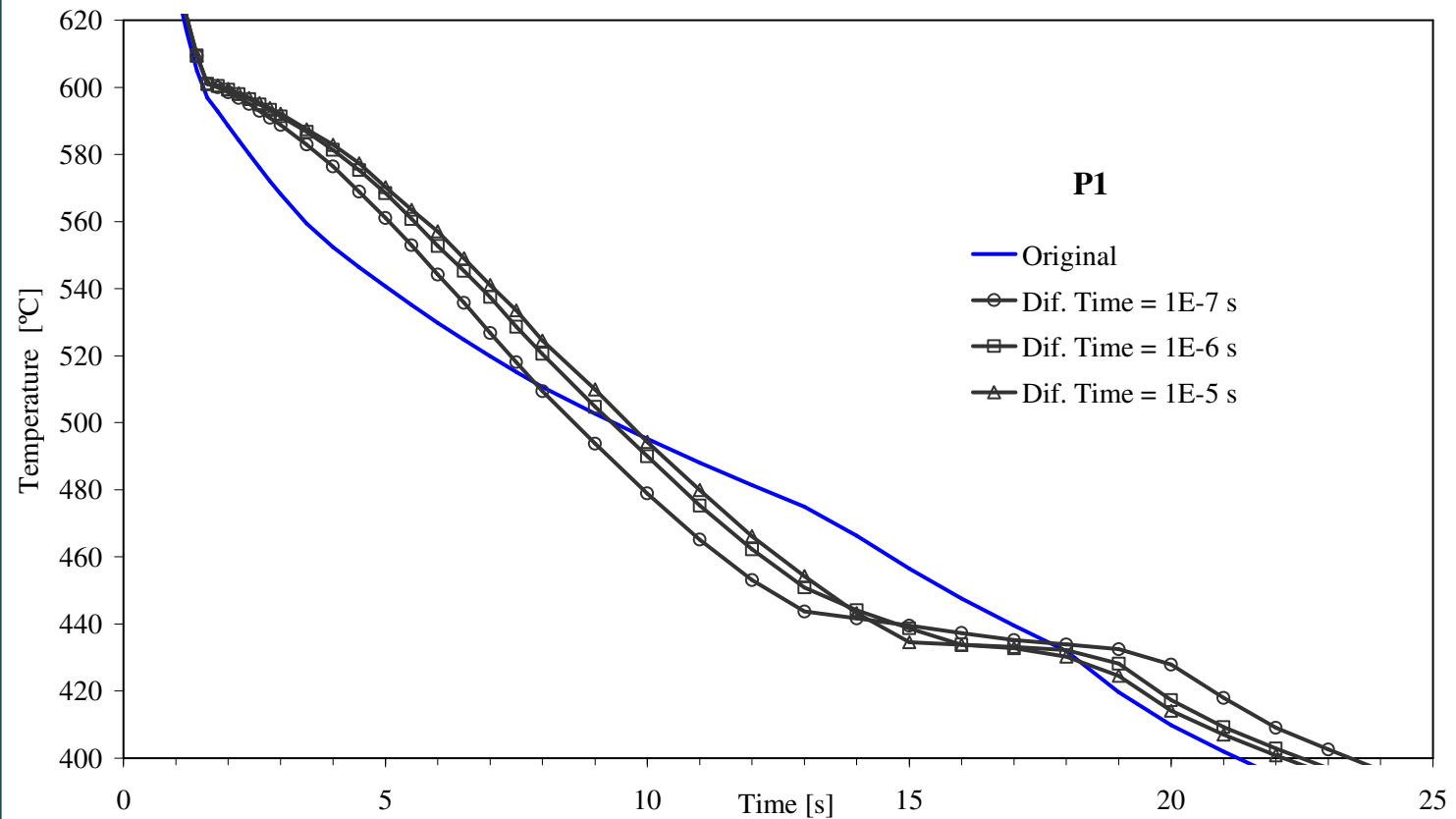
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Comparing cooling curves before and after calculating energy release with ChemApp in function of diffusion time



Diffusion length → 1E-7 m

Comparing calculated cooling curves to experimental casting curve

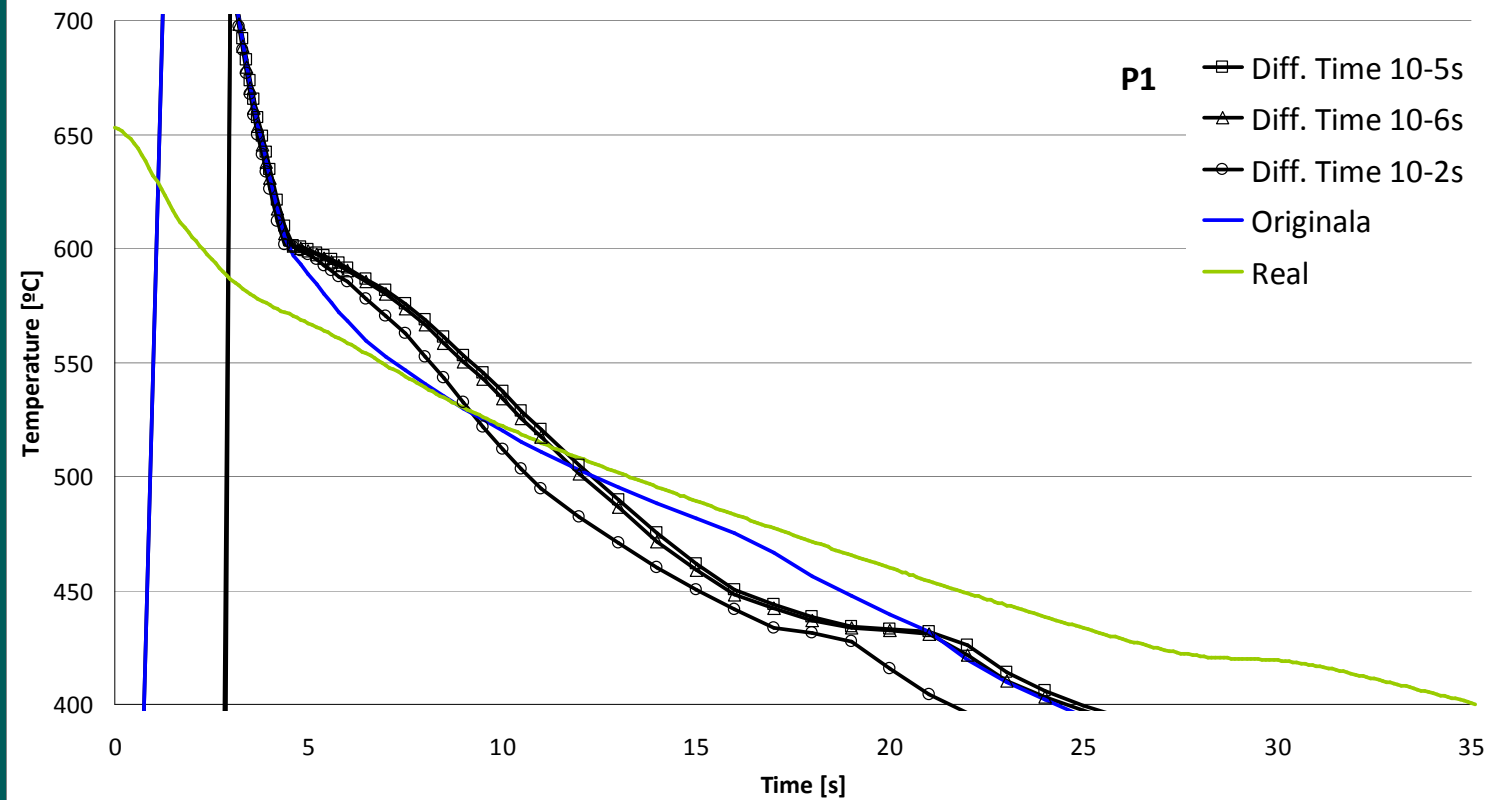
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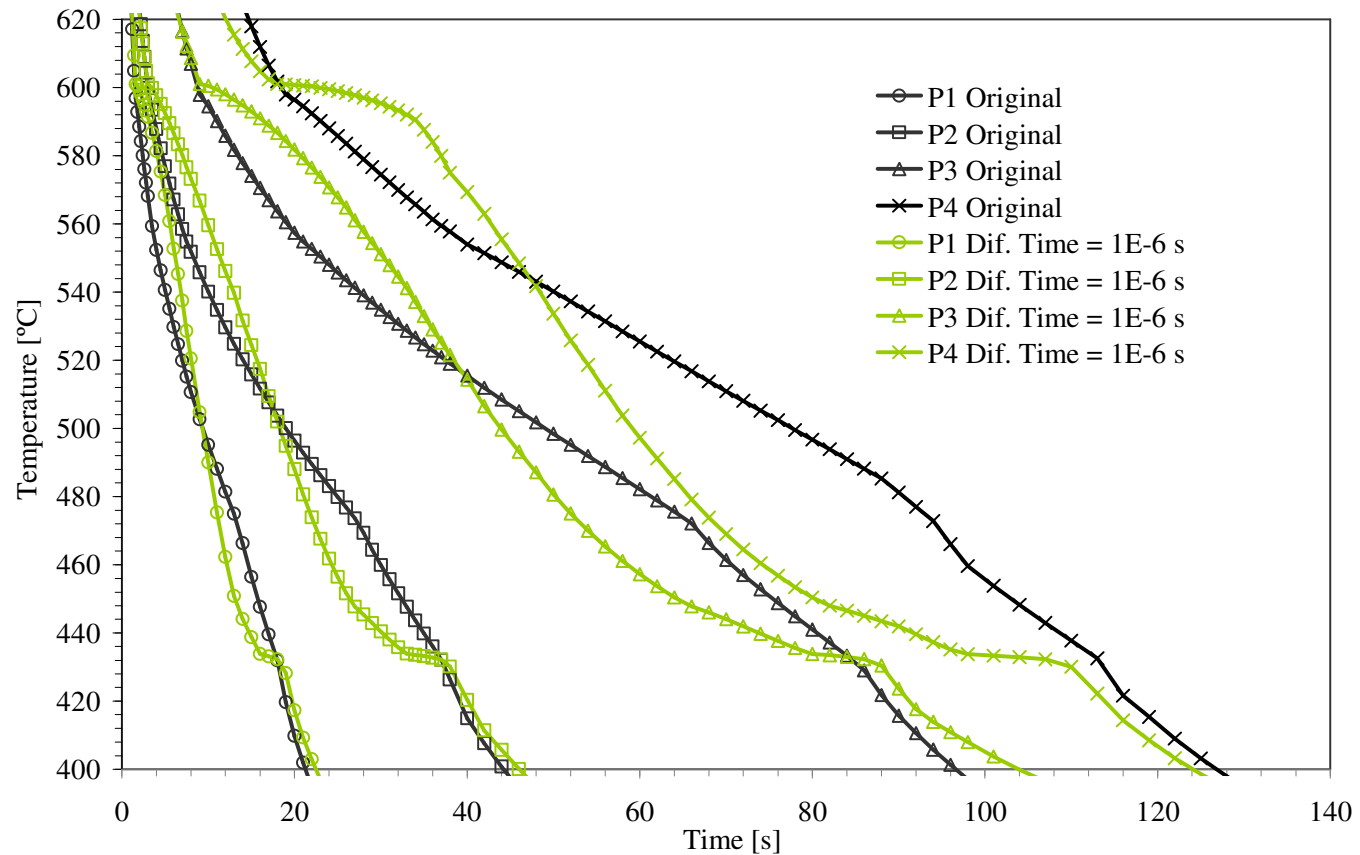
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Comparing cooling curves before and after calculating energy release with ChemApp in function of cooling rate



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Diffusion length → 1E-7 m

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SUMMARY AND FUTURE WORK

- A microstructure growth model has been modelled, supported by Calphad-ChemApp calculations to handle multi-component alloys.
- ChemApp calculates the appearance of phases and their temperature path. Released heat is also computed.
- Unlike diffusion times during calculations result in different amount of the solid phases and heat release, approaching to Scheil solidification (perfect solute mixture in liquid) as the diffusion time increases.
- The temperature where solidification finishes also varies with changing the diffusion time.
- Alloying element segregation is also computed in function of diffusion time.
- It is proved that using Calphad based tools, it is no longer accepted that solid fraction and latent heat release are proportional.

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CONCLUSIONS AND FUTURE WORK

- Casting simulations carried out with WinCast using ChemApp provided heat release data, improve the resulting calculation in comparison to DSC data for latent heat.
- A better description of the primary α -Mg phase and eutectic transformation is also obtained.
- **Future work:**
 - To achieve full coupling of the micro-model and ChemApp calculations to WinCast.
 - Improve calculations to approach the calculated curves to the experimental ones.

Thank You For your Attention !!

Danke Für Ihre Aufmerksamkeit !!